

**Possible superconductivity in Fe-Sb based materials: Density functional study of LiFeSb**Lijun Zhang,<sup>1</sup> Alaska Subedi,<sup>1,2</sup> D. J. Singh,<sup>1</sup> and M. H. Du<sup>1</sup><sup>1</sup>*Materials Science and Technology Division, Oak Ridge National Laboratory, Oak Ridge, Tennessee 37831-6114, USA*<sup>2</sup>*Department of Physics and Astronomy, University of Tennessee, Knoxville, Tennessee 37996, USA*

(Received 19 August 2008; revised manuscript received 27 October 2008; published 19 November 2008)

We investigate the electronic and other properties of the hypothetical compound LiFeSb in relation to superconducting LiFeAs and FeSe using density-functional calculations. The results show that LiFeSb in the LiFeAs structure would be dynamically stable in the sense of having no unstable phonon modes and would have very similar electronic and magnetic properties to the layered Fe-based superconductors. Importantly, a very similar structure for the Fermi surface and a spin-density wave related to but stronger than that in the corresponding As compound is found. These results are indicative of possible superconductivity analogous to the Fe-As based compounds if the spin-density wave can be suppressed by doping or other means. Prospects for synthesizing this material in pure form or in solid solution with FeTe are discussed.

DOI: [10.1103/PhysRevB.78.174520](https://doi.org/10.1103/PhysRevB.78.174520)

PACS number(s): 74.25.Jb, 74.70.Dd, 71.18.+y, 74.25.Kc

The finding of high-temperature superconductivity ( $T_c \sim 26$  K) in electron-doped LaFeAsO<sub>1-x</sub>F<sub>x</sub> (Ref. 1) has resulted in widespread interest and exploration of related materials, some of which have  $T_c$  exceeding 55 K. In particular, superconductivity has been found in iron based oxyarsenides by replacing La with other rare-earth metals,<sup>2-6</sup> as well as oxygen-free arsenides such as doped BaFe<sub>2</sub>As<sub>2</sub>,<sup>7,8</sup> SrFe<sub>2</sub>As<sub>2</sub>,<sup>9</sup> CaFe<sub>2</sub>As<sub>2</sub>,<sup>10,11</sup> and LiFeAs.<sup>12-14</sup> The common structural feature of this family of materials is the appearance of Fe-As layers. These consist of an Fe square planar sheet tetrahedrally coordinated by As atoms from above and below. In addition, superconductivity occurs in doped LaFePO,<sup>15-17</sup> although with a lower  $T_c$  and in PbO structure  $\alpha$ -FeSe<sub>1-x</sub>.<sup>18-20</sup> These latter compounds also feature an Fe square lattice and a tetrahedral coordination of the Fe, though not with As. Importantly, the critical temperature of FeSe<sub>1-x</sub> increases strongly with either Te substitution<sup>19</sup> or pressure, reaching 27 K.<sup>20</sup> This high value of  $T_c$  under pressure implies a relationship with the Fe-As superconductors, which is also supported by similarities of the properties and theoretical studies.<sup>21</sup> At present there is strong interest in finding new high-temperature Fe-based superconductors and especially in finding materials with higher critical temperature.

One obvious direction is to examine antimonides. This is motivated by the fact that the properties of LaFePO and LaFeAsO appear to be closely related, suggesting a similar mechanism of superconductivity, and furthermore the compound with the heavier pnictogen (As) has the higher  $T_c$  when doped. However, this is highly nontrivial from a chemical perspective because Sb has a strong tendency to form Sb-Sb bonds in compounds. This leads to a strong tendency for transition-metal compounds of Sb to contain more Sb than transition metal, as, for example, in skutterudite CoSb<sub>3</sub> and LaFe<sub>4</sub>Sb<sub>12</sub> or marcasite structure FeSb<sub>2</sub>, although FeSb is a known phase.<sup>22</sup> One way forward is provided by noting the structural similarity of LiFeAs with PbO structure, FeSe<sub>1-x</sub> and FeTe<sub>1-x</sub>. The chalcogenides, whose chemical formulas should more correctly be written as Fe<sub>1+x</sub>Se and Fe<sub>1+x</sub>Te, occur in a tetragonal structure with space group  $P4/nmm$  similar to LiFeAs and consist of an Fe square lattice tetrahedrally coordinated with Se/Te ions, the same as in the structure of the Fe-As superconductors.<sup>23-26</sup> These chal-

cogenides form with excess Fe, which occurs in a partially filled 2c site, in particular the cation site forming formally an enlarged tetrahedron around the Fe and approximately five-fold coordinated by Te.<sup>23,25,26</sup> This is the same site that is occupied by Li in LiFeAs. Therefore, there is a close structural similarity between LiFeAs and the chalcogenides Fe<sub>1+x</sub>Se and Fe<sub>1+x</sub>Te. In particular the structure of LiFeAs is obtained by allowing full filling of the 2c cation site with Li<sup>+</sup> and replacement of Te<sup>2-</sup> by As<sup>3-</sup>. Therefore we focus on hypothetical LiFeSb since it may be possible to form it or at the very least some range of solid solution between Fe<sub>1+x</sub>Te and LiFeSb should be experimentally accessible, especially considering that alloys of related phases containing Te and Sb typically form as in, e.g., the Bi-Sb-Te, AgSbTe-PbTe, and AgSbTe-GeTe thermoelectrics and also that there are many known Zintl-type phases based on Li, Sb, and metal atoms.

The crystal structure of LiFeSb is assumed to be isostructural with LiFeAs with the space group of  $P4/nmm$ .<sup>12-14</sup> As shown in Fig. 1, the Fe-Sb layers formed by edge-shared tetrahedral FeSb<sub>4</sub> units are alternately spaced along the *c*-axis direction and intercalated with Li. Note that Li, which occurs as Li<sup>+</sup>, is coordinated by Sb. The structural parameters were calculated by local-density approximation (LDA) total-energy minimization with the full-potential linearized augmented plane-wave (LAPW) method.<sup>27</sup> The calculated tetragonal lattice parameters are  $a=4.0351$  Å,  $c=6.3712$  Å, and internal coordinates Li(2c) (0.25,0.25,0.697), Fe(2a) (0.75,0.25,0), and Sb(2c) (0.25,0.25,0.228). The Fe-Sb bond length is 2.486 Å, slightly larger than 2.4204 Å for LiFeAs,<sup>14</sup> which might be attributed to the larger size of the Sb<sup>3-</sup> anion relative to As<sup>3-</sup>. The Fe-Fe distance is 2.853 Å, also a bit larger than the corresponding value (2.6809 Å) in LiFeAs but still short enough for direct Fe-Fe interaction.

The electronic structure and magnetic property calculations were performed within LDA-LAPW method, similar to previous reports.<sup>21,28,29</sup> LAPW sphere radii of  $1.8a_0$ ,  $2.0a_0$ , and  $2.1a_0$  were used for Li, Fe, and Sb, respectively. Converged basis sets were used. These consisted of LAPW functions with a plane-wave cutoff determined as  $R_{Li}k_{max}=8.0$  plus local orbitals both to relax linearization and to include the semicore states. The zone sampling for the self-consistent

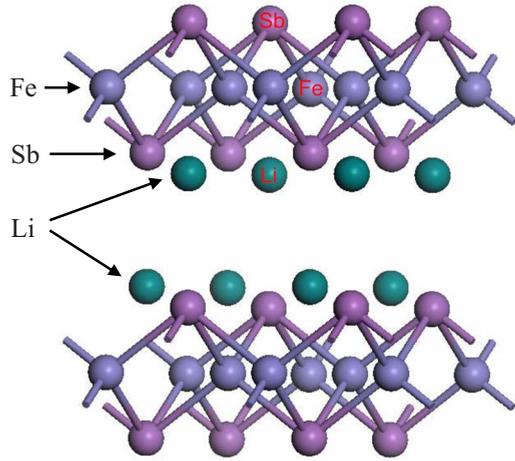


FIG. 1. (Color online) Crystal structure of hypothetical LiFeSb with the relaxed structural parameters from LAPW-LDA total-energy minimization. Note that while for clarity similar size spheres are used for the different atoms, from a crystal chemical point of view  $\text{Sb}^{3-}$  anions are very large while  $\text{Li}^+$  is very small.

calculations was done using the special  $\mathbf{k}$ -point method, with a  $16 \times 16 \times 8$  grid. Finer grids were used for the density of states (DOS) and Fermi surface. The lattice-dynamical properties were calculated through the frozen phonon method<sup>30</sup> (or small displacement method<sup>31</sup>). The required forces were obtained through the projector augmented-wave (PAW) method<sup>32</sup> in VASP code within the generalized gradient approximation (GGA) of Perdew, Burke, and Ernzerhof (PBE).<sup>33</sup> We also fully relaxed crystal structure and calculated electronic structures with PBE-PAW method and the results show excellent agreement with those by LDA-LAPW method (with a remarkably small maximum discrepancy of 0.3% in structural parameters). This cross-checking supports the reliability of the calculations and consistency of the different methods employed. We emphasize that these calculations for the structure were done without the inclusion of magnetism.

As discussed in detail in Ref. 34, there is generally a substantial underestimation of the pnictogen heights in these compounds when calculations are done in this way, while on the other hand in magnetic calculations LDA and GGA results differ, with the GGA giving much larger magnetic moments than experiment as well as magnetism that persists throughout the phase diagram in disagreement with experiment. LDA calculations done with the GGA structure give an intermediate state less magnetic than the GGA calculations but more magnetic than experiment, while in magnetic LDA calculations the As height is still substantially underestimated. In fact, LDA calculations done at the nonmagnetic LDA As height (which agrees with the GGA As height) give the weakest magnetism, closest to experiment, though still overestimating the strength of the SDW. It was conjectured that these problems are consequences of strong spin fluctuations.

This conjecture is supported both by comparison of theoretical results with experiment as well as experimental observations, such as the highly unusual increasing with  $T$  susceptibility,  $\chi(T)$  above the spin-density wave (SDW) ordering

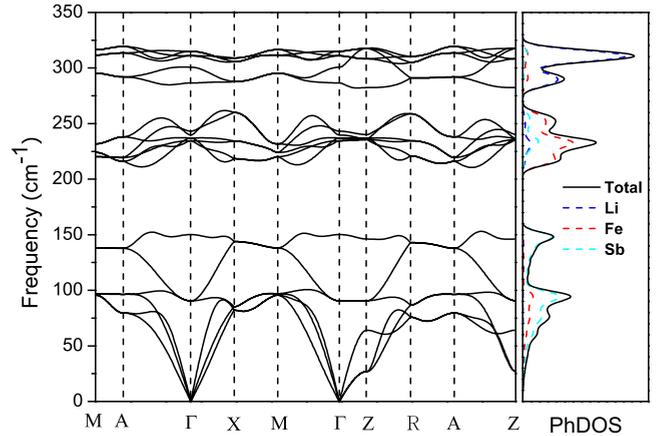


FIG. 2. (Color online) Left panel: calculated phonon-dispersion curves for LiFeSb. Right panel: the total and projected (onto atoms) phonon DOSs.

temperature observed in some compounds.<sup>35–37</sup> Such an increasing shape suggests that strong magnetic correlations with the character of the SDW persist well above  $T_N$ . Since this itinerant magnetic state is driven by electrons at the Fermi energy a related electronic reconstruction of the Fermi surface may also be expected above  $T_N$ ; this may be seen perhaps in photoemission. This shape of  $\chi(T)$  persists also in the normal state for doped samples, where there is superconductivity but no SDW.<sup>37</sup> Other evidence for strong spin fluctuations comes from core-level spectroscopy<sup>38</sup> and transport data showing strong scattering above the ordering temperature.<sup>35</sup> Returning to the increasing  $\chi(T)$ , which continues up to high temperature, one interesting possibility is that this represents preformed pairs that can condense into either superconducting or SDW order and which begin forming at a very high nonobserved temperature above which  $\chi(T)$  would return to a more normal decreasing with  $T$  shape. In any case, in this work where we compare the compounds, we consistently used the relaxed atomic coordinates from nonmagnetic calculations.

A requirement for a compound to be made is that the lattice be stable. We verified that this is the case for hypothetical LiFeSb by calculating the vibrational modes of the compound.<sup>39</sup> We find no soft or unstable modes and no soft elastic constants. The calculated phonon-dispersion curve and phonon DOS for LiFeSb are shown in Fig. 2. Due to larger difference in atomic weights compared to LiFeAs the phonon spectrum of LiFeSb is divided into three separated manifolds. The region of high frequencies (above  $275 \text{ cm}^{-1}$ ) is dominated by Li, while the moderate (between 200 and  $275 \text{ cm}^{-1}$ ) and low (below  $150 \text{ cm}^{-1}$ ) frequency manifolds mainly derive from Fe and Sb, respectively. As may be seen, all the phonon frequencies are safely positive and there are no optical phonon branches with dispersions that dip toward zero frequency. This shows that the  $P4/nmm$  structure of LiFeSb is dynamically stable. We also calculated the heat of formation from the elements. We obtain  $-0.51 \text{ eV/f.u.}$  (i.e.,  $-49 \text{ kJ/mol f.u.}$ ), which indicates that the compound may be delicate but would at least be stable against decomposition into elements. Therefore we continue to discuss the magnetic and electronic properties.

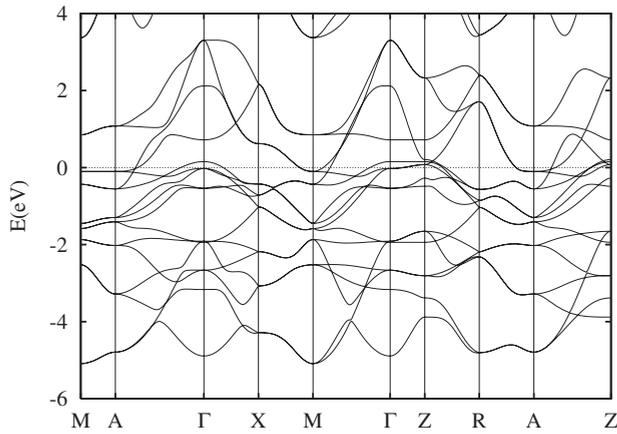


FIG. 3. Calculated LDA band structure of LiFeSb using the calculated structural parameters. The Fermi energy is at 0 eV.

Our main results for the electronic structure of LiFeSb are given in Figs. 3–5, which show the calculated band structure, electronic DOS, and Fermi surface, respectively. The general shape of band structure near the Fermi energy  $E_F$  is very similar to the calculated results for LiFeAs.<sup>29,40</sup> There are compensating heavy hole and electron Fermi surfaces, with two electron cylinders at the zone corner ( $M$ ) and hole surfaces around the zone center. The hole surfaces consist of two-dimensional (2D) cylindrical and small heavy three-dimensional (3D) sections. Similar to the Fe-As based materials,<sup>28,29,40–43</sup> the electron Fermi surface of LiFeSb may be described as two intersecting cylindrical sections of elliptical cross section, with major axes at  $90^\circ$  to each other and centered at the  $M$  point. We find somewhat a different hole Fermi-surface structure from LiFeAs, with only one complete hole cylinder at the zone center, along with two additional heavier 3D hole pockets. It can be seen that electron cylinders are more two dimensional than in LiFeAs. Also, the 2D hole cylinder is close in size to that of the electron cylinders, which may be expected to lead to nesting. Thus there is strong nesting of Fermi surface at the 2D nesting

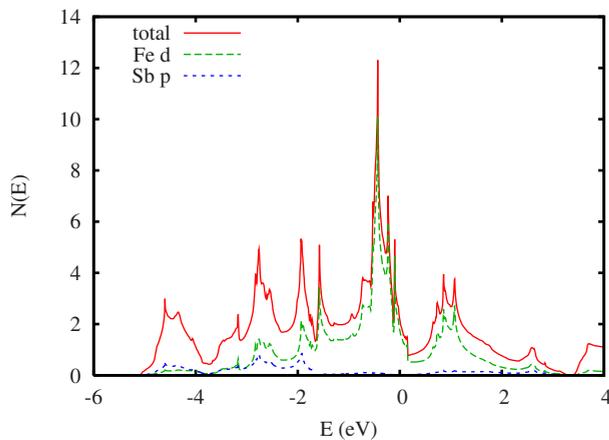


FIG. 4. (Color online) Calculated total and partial electronic DOSs for LiFeSb on a per f.u. basis. The contribution from Li  $2s$  state lying in deep energy range was not shown. The projections are onto the LAPW spheres, thus the Sb  $5p$  contribution is underestimated owing to its more extended orbitals.

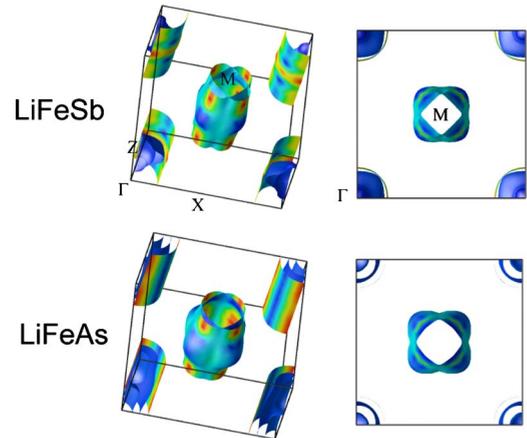


FIG. 5. (Color online) Calculated LDA Fermi surface of LiFeSb in comparison with LiFeAs, shaded by band velocity with blue as low velocity. The right panels are top views along the  $c$ -axis direction.

vector  $(\pi, \pi)$ . This would be expected to lead to an SDW state related to the  $M$  point, as in the Fe-As based superconductors.<sup>42–47</sup> We studied the energetic stability of SDW state for LiFeSb directly using a doubled cell containing lines of Fe atoms with parallel spin in the Fe-Sb layers and do, in fact, find a stable SDW state. Within the LDA with the LDA structural parameters, the local spin moment of the SDW state is  $1.12\mu_B$ , much larger than the corresponding value ( $0.69\mu_B$ ) for LiFeAs calculated in the same way,<sup>29</sup> indicating that LiFeSb has a more stable SDW.

The qualitative similarity to the electronic structure of the Fe-As based superconductors<sup>28,29,40–43</sup> is also evident in the DOS. The Sb  $p$  states are located mainly below  $-1.7$  eV relative to  $E_F$  and are only moderately hybridized with the Fe  $d$  states, indicating that Sb is anionic with valence close to 3. The DOS near the Fermi level is dominated by Fe  $d$  states, deriving from the metallic  $\text{Fe}^{2+}$  sublattice with direct Fe-Fe interactions, and has a characteristic pseudogap near  $E_F$ . In fact,  $E_F$  lies on the low energy side of the pseudogap, where  $N(E_F)$  is decreasing with energy but still high. Specifically, the value of  $N(E_F)=2.2$   $\text{eV}^{-1}$  per Fe, both spins, is much larger than that for LiFeAs and is comparable to the oxyarsenides [e.g.,  $N(E_F)$  calculated in the same way for LaFeAsO is  $2.6$   $\text{eV}^{-1}$ ], which are the Fe-As compounds with higher  $T_c$ . For comparison, the values for LiFeAs and  $\text{BaFe}_2\text{As}_2$  are  $1.79$  and  $1.53$   $\text{eV}^{-1}$  on a per Fe basis, respectively, when calculated in the same way.<sup>29</sup>

Within the Stoner theory, the appearance of an instability of the paramagnetic state toward itinerant ferromagnetism would be determined by the criterion  $N(E_F)I > 1$ , where  $I$  is the Stoner parameter, with the typical value in Fe compounds of  $I \sim 0.7\text{--}0.8$  eV. Thus, the significantly higher  $N(E_F)$  in LiFeSb would inevitably place it closer to magnetism in general than LiFeAs or  $\text{BaFe}_2\text{As}_2$ . While the mechanism of superconductivity has yet to be established, there is accumulating evidence of a connection with magnetism, and so chemically tuning the proximity to magnetism is a likely strategy for modifying the superconductivity. In general, the Fe-based superconductors exhibit temperature-induced mag-

netic and structural phase transitions with a SDW character related to the Fermi-surface nesting.<sup>7,48,49</sup> Superconductivity appears as the spin-density wave is suppressed by doping or pressure.

Electronic structure calculations<sup>28,29,40–43</sup> show that all these materials have compensating small electron and hole Fermi surfaces, with nesting between 2D electron sheets and heavier 2D hole sheets, which are separated by  $(\pi, \pi)$ . This is associated with the SDW magnetic state.<sup>42–47</sup>

Within this framework, the idea that going to heavier ligands may be beneficial for superconductivity is supported by previous density-functional calculations. These have shown that the electronic structures of the Fe-As superconductors are rather ionic with the exception of the Fe layers, which are metallic due mainly to Fe-Fe interactions.<sup>28</sup> This is different from the cuprates where hopping is through the O atoms in the CuO<sub>2</sub> planes and implies that the ligand (O/As) atoms play a less crucial role in the properties of the FeAs superconductors than in the cuprates. Furthermore, it has been found that there is a strong connection between the As position above the Fe plane and the magnetic properties, with higher positions yielding stronger magnetism.<sup>34,44</sup> This is supported by calculations comparing FeSe and FeTe (Ref. 21) and for hypothetical LaFeSbO in comparison with LaFeAsO.<sup>50</sup> In both cases stronger magnetism is found in going to the larger ligand, which because of its size is then further from the Fe plane yielding narrower bands and higher  $N(E_F)$ . The combination of a stronger SDW and higher  $N(E_F)$  leading to stronger spin fluctuations in general and in particular away from the nesting vector may be crucial. This is because the ordered SDW is antagonistic to superconductivity. In scenarios where the associated spin fluctuations that couple the electron and hole Fermi-surface sections play the main role in pairing, as discussed in Refs. 46 and 51, spin fluctuations away from the nesting vector while not directly pairing may play a very important role. This is because they would compete with the SDW preventing long-range order and leading to a renormalized paramagnetic state even though in mean field the SDW may be the predicted ground state as in the LDA. In any case, in these scenarios the role of doping is to weaken and broaden the peak in the suscep-

tibility associated with the nesting, destroying the SDW in favor of a state with spin fluctuations around the zone corner. We also note that in case the SDW is not destroyed by doping alone; it may be possible for it to be destroyed by disorder yielding superconductivity in an alloy system such as Fe<sub>1+x</sub>Te-LiFeSb. This may be possible because the SDW is related to a divergence in the peak of the susceptibility,  $\chi(\mathbf{q})$ , while within a spin-fluctuation mediated framework superconductivity will in general be related not to the peak value but to an integral over the Fermi surface, i.e., a high average value over some region of the zone, specifically the region for which the wave vector connects the electron and hole Fermi surfaces.<sup>46,52</sup> Also near divergences will be pair breaking for superconductivity. This may also partly explain why superconductivity in these phases is relatively robust against alloying with Zn or Co in the Fe planes,<sup>53–55</sup> even though in an unconventional superconductor scattering, including non-magnetic scattering, is pair breaking. Thus, even if the magnetic ground state cannot be destroyed by doping in Fe<sub>1+x</sub>Te or LiFeSb (supposing that this can be synthesized) it may be destroyed in favor of superconductivity in the solid solution between these two compounds.

In any case, our results show that, if it can be synthesized, LiFeSb will have electronic and magnetic properties closely related to those of the Fe-As based superconductors and in particular will show a qualitatively similar Fermi-surface structure and tendency toward an SDW state. In comparison with LiFeAs, it will have a higher  $N(E_F)$  and a stronger SDW. This may favor higher critical temperatures at least within a scenario with interband pairing mediated by spin fluctuations associated with the Fermi-surface nesting. In addition, this material is found to be dynamically stable, evidenced by the absence of any unstable phonon modes. It is also worth noting that this solid solution contains no elements as toxic as As. As such it would be of considerable interest to attempt synthesis of this compound or its solid solution with Fe<sub>1+x</sub>Se, LiFeAs, and especially Fe<sub>1+x</sub>Te.

We are grateful for helpful discussions with D. Mandrus, I. I. Mazin, and B. C. Sales. This work was supported by the Department of Energy, Division of Materials Sciences and Engineering.

<sup>1</sup>Y. Kamihara, T. Watanabe, M. Hirano, and H. Hosono, *J. Am. Chem. Soc.* **130**, 3296 (2008).

<sup>2</sup>C. Wang *et al.*, *Europhys. Lett.* **83**, 67006 (2008).

<sup>3</sup>Z.-A. Ren *et al.*, *Europhys. Lett.* **83**, 17002 (2008).

<sup>4</sup>A. S. Sefat, M. A. McGuire, B. C. Sales, R. Jin, J. Y. Howe, and D. Mandrus, *Phys. Rev. B* **77**, 174503 (2008).

<sup>5</sup>G. F. Chen, Z. Li, D. Wu, G. Li, W. Z. Hu, J. Dong, P. Zheng, J. L. Luo, and N. L. Wang, *Phys. Rev. Lett.* **100**, 247002 (2008).

<sup>6</sup>H. Wen, G. Mu, L. Fang, H. Yang, and X. Zhu, *Europhys. Lett.* **82**, 17009 (2008).

<sup>7</sup>M. Rotter, M. Tegel, D. Johrendt, I. Schellenberg, W. Hermes, and R. Pottgen, *Phys. Rev. B* **78**, 020503(R) (2008).

<sup>8</sup>M. Rotter, M. Tegel, and D. Johrendt, *Phys. Rev. Lett.* **101**, 107006 (2008).

<sup>9</sup>G. F. Chen, Z. Li, G. Li, W. Z. Hu, J. Dong, X. D. Zhang, P. Zheng, N. L. Wang, and J. L. Luo, *Chin. Phys. Lett.* **25**, 3403 (2008).

<sup>10</sup>N. Ni, S. Nandi, A. Kreyssig, A. I. Goldman, E. D. Mun, S. L. Bud'ko, and P. C. Canfield, *Phys. Rev. B* **78**, 014523 (2008).

<sup>11</sup>M. S. Torikachvili, S. L. Bud'ko, N. Ni, and P. C. Canfield, *Phys. Rev. Lett.* **101**, 057006 (2008).

<sup>12</sup>X. C. Wang, Q. Q. Liu, Y. X. Lv, W. B. Gao, L. X. Yang, R. C. Yu, F. Y. Li, and C. Q. Jin, arXiv:0806.4688

<sup>13</sup>M. J. Pitcher, D. R. Parker, P. Adamson, S. J. C. Herkelrath, A. T. Boothroyd, and S. J. Clarke, *Chem. Commun. (Cambridge)* (to be published).

<sup>14</sup>J. H. Tapp, Z. Tang, B. Lv, K. Sasmal, B. Lorenz, P. C. W. Chu, and A. M. Guloy, *Phys. Rev. B* **78**, 060505(R) (2008).

- <sup>15</sup>Y. Kamihara, H. Hiramatsu, M. Hirano, R. Kawamura, H. Yanagi, T. Kamiya, and H. Hosono, *J. Am. Chem. Soc.* **128**, 10012 (2006).
- <sup>16</sup>C. Y. Liang, R. C. Che, H. X. Yang, H. F. Tian, R. J. Xiao, J. B. Lu, R. Li, and J. Q. Li, *Supercond. Sci. Technol.* **20**, 687 (2007).
- <sup>17</sup>T. M. McQueen, M. Regulacio, A. J. Williams, Q. Huang, J. W. Lynn, Y. S. Hor, D. V. West, M. A. Green, and R. J. Cava, *Phys. Rev. B* **78**, 024521 (2008).
- <sup>18</sup>F.-C. Hsu *et al.*, *Proc. Natl. Acad. Sci. U.S.A.* **105**, 14262 (2008).
- <sup>19</sup>K.-W. Yeh *et al.*, arXiv:0808.0474 (unpublished).
- <sup>20</sup>Y. Mizuguchi, F. Tomioka, S. Tsuda, T. Yamaguchi, and Y. Takano, *Appl. Phys. Lett.* **93**, 152505 (2008).
- <sup>21</sup>A. Subedi, L. Zhang, D. J. Singh, and M.-H. Du, *Phys. Rev. B* **78**, 134514 (2008).
- <sup>22</sup>A. Kjekshus and K. P. Walseth, *Acta Chem. Scand.* (1947-1973) **23**, 2621 (1969).
- <sup>23</sup>F. Gronvold and H. Haraldsen, *Acta Chem. Scand.* (1947-1973) **8**, 1927 (1954).
- <sup>24</sup>S. Chiba, *J. Phys. Soc. Jpn.* **10**, 837 (1955).
- <sup>25</sup>D. M. Finlayson, D. Greig, J. P. Llewellyn, and T. Smith, *Proc. Phys. Soc. London, Sect. B* **69**, 860 (1956).
- <sup>26</sup>J. Leciejewicz, *Acta Chem. Scand.* (1947-1973) **17**, 2593 (1963).
- <sup>27</sup>D. J. Singh and L. Nordstrom, *Planewaves Pseudopotentials and the LAPW Method*, 2nd ed. (Springer, Berlin, 2006).
- <sup>28</sup>D. J. Singh and M.-H. Du, *Phys. Rev. Lett.* **100**, 237003 (2008).
- <sup>29</sup>D. J. Singh, *Phys. Rev. B* **78**, 094511 (2008).
- <sup>30</sup>A. Togo, FROPHO: Frozen Phonon analyzer for periodic boundary-condition materials.
- <sup>31</sup>D. Alfè, G. D. Price, and M. J. Gillan, *Phys. Rev. B* **64**, 045123 (2001).
- <sup>32</sup>G. Kresse and D. Joubert, *Phys. Rev. B* **59**, 1758 (1999).
- <sup>33</sup>J. P. Perdew, K. Burke, and M. Ernzerhof, *Phys. Rev. Lett.* **77**, 3865 (1996).
- <sup>34</sup>I. I. Mazin, M. D. Johannes, L. Boeri, K. Koepernik, and D. J. Singh, *Phys. Rev. B* **78**, 085104 (2008).
- <sup>35</sup>M. A. McGuire *et al.*, *Phys. Rev. B* **78**, 094517 (2008).
- <sup>36</sup>H. H. Klauss *et al.*, *Phys. Rev. Lett.* **101**, 077005 (2008).
- <sup>37</sup>R. Klingeler, N. Leps, I. Hellmann, A. Popa, A. Kondrat, J. Hamann-Borrero, G. Behr, V. Kataev, and B. Buechner, arXiv:0808.0708 (unpublished).
- <sup>38</sup>F. Bondino *et al.*, arXiv:0807.3781 (unpublished).
- <sup>39</sup>S. Baroni, S. de Gironcoli, A. Dal Corso, and P. Giannozzi, *Rev. Mod. Phys.* **73**, 515 (2001).
- <sup>40</sup>I. A. Nekrasov, Z. V. Pchelkina, and M. V. Sadovskii, arXiv:0807.1010 (unpublished).
- <sup>41</sup>I. A. Nekrasov, Z. V. Pchelkina, and M. V. Sadovskii, *JETP Lett.* **88**, 144 (2008).
- <sup>42</sup>F. Ma and Z.-Y. Lu, *Phys. Rev. B* **78**, 033111 (2008).
- <sup>43</sup>F. Ma, Z.-Y. Lu, and T. Xiang, arXiv:0806.3526 (unpublished).
- <sup>44</sup>Z. P. Yin, S. Lebegue, M. J. Han, B. Neal, S. Y. Savrasov, and W. E. Pickett, *Phys. Rev. Lett.* **101**, 047001 (2008).
- <sup>45</sup>J. Dong *et al.*, *Europhys. Lett.* **83**, 27006 (2008).
- <sup>46</sup>I. I. Mazin, D. J. Singh, M. D. Johannes, and M.-H. Du, *Phys. Rev. Lett.* **101**, 057003 (2008).
- <sup>47</sup>T. Yildirim, *Phys. Rev. Lett.* **101**, 057010 (2008).
- <sup>48</sup>C. de la Cruz *et al.*, *Nature (London)* **453**, 899 (2008).
- <sup>49</sup>T. Nomura, S. W. Kim, Y. Kamihara, M. Hirano, P. V. Sushko, K. Kato, M. Takata, A. L. Shluger, and H. Hosono, arXiv:0804.3569 (unpublished).
- <sup>50</sup>C.-Y. Moon, S. Y. Park, and H. J. Choi, arXiv:0808.1348 (unpublished).
- <sup>51</sup>K. Kuroki, S. Onari, R. Arita, H. Usui, Y. Tanaka, H. Kontani, and H. Aoki, *Phys. Rev. Lett.* **101**, 087004 (2008).
- <sup>52</sup>D. Fay and J. Appel, *Phys. Rev. B* **22**, 3173 (1980).
- <sup>53</sup>A. S. Sefat, A. Huq, M. A. McGuire, R. Jin, B. C. Sales, D. Mandrus, L. M. D. Cranswick, P. W. Stephens, and K. H. Stone, *Phys. Rev. B* **78**, 104505 (2008).
- <sup>54</sup>A. S. Sefat, R. Jin, M. A. McGuire, B. C. Sales, D. J. Singh, and D. Mandrus, *Phys. Rev. Lett.* **101**, 117004 (2008).
- <sup>55</sup>Y. K. Li, X. Lin, C. Wang, L. J. Li, Z. W. Zhu, M. H. Q. Tao, Q. B. Wang, G. H. Cao, and Z. A. Xu, arXiv:0808.0328 (unpublished).